In-situ XRD during sputtering of Shape Memory Alloy (SMA) Ni-Ti thin films



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Abstract: Ni-Ti thin films present great advantages for the fabrication of micro-actuators [1]. Its structural evolution has been followed *in situ* by XRD during sputter deposition and during annealing. The series of first experiments on *in-situ* sputtering (Ni-Ti) and co-sputtering (Ni-Ti + Ti) led to a much better understanding of the structural development of the Ni-Ti films and its correlation with the corresponding deposition parameters.

The sequence of the phases formation and their preferential orientations had been followed for different deposition conditions and substrate type.

Experimental Setup

A sputter deposition chamber for the *in-situ* study of film growth by synchrotron x-ray diffraction and reflectivity was used [2]. The two magnetrons are placed at a distance of 100 mm from the substrates and tilted 30 degrees away from the substrate normal. To avoid cross contamination of the targets, chimneys (1 inch long) are mounted on the magnetrons. Air–pressure-controlled shutters are placed in front of the chimneys. Substrate heating up to 650 °C and different substrate bias voltage are possible. All the depositions were carried out with substrate heating at 470°C. Annealing after deposition was also at 470°C, except for sample 13 where 545° C annealing was also used.





Thin films production

The properties of Ni-Ti thin films are strongly dependent on the chemical composition. The chemical composition of Ni-Ti sputtered films may deviate by up to 2 at% (lower Ti) when compared to the target composition [3]; this deviation gives rise to lower transformation temperature ranges. In order to ensure transformation temperatures above room temperature, sputtering using multiple targets is an obvious solution.

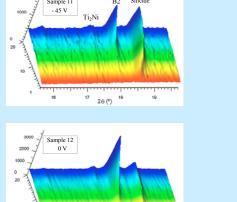
At ROBL (BM20), experimental *in situ* studies of the structural evolution during sputtering (using a single target Ni-Ti) and co-sputtering (using Ni-Ti and Ti targets) were carried out.

Parameters used for sputtering deposition

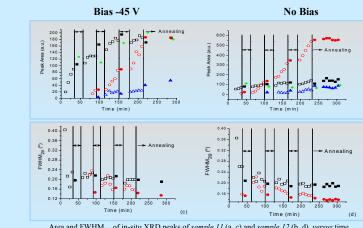
<u>Ni-11</u>							$N_{1}-11+11$						
Sample	Power (W)	Substrate	Bias (V)	Deposition (min)	Annealing (min)	Sample	Power (W)		Substrate	rate Bias	Deposition procedure	Annealing	
2	40	SiO2/Si (100)	-45	154	90		Ni-Ti	Ti		(1)		(min)	
3	40	Si (100)	0	139	70	11	40	5	Si (100)	-45	4 × (35 min Ni-Ti + Ti	85	
4	40	Si (100)	-45	120	62	12	40	5	Si (100)	0	co-sputtering / 25 min annealing)	90	
<u> </u>	-			.20		12	00	-	0: (100)	0	1h 30 min (Ni-Ti + Ti	46 at 470°C	
5	40	SiO2/Si (100)	0	124	98	13	80	5	Si (100)	0	co-sputtering)	100 at 545°C	

Experimental results

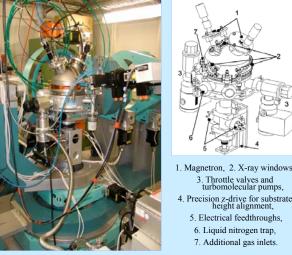
Co-sputtering using Ni-Ti + Ti targets

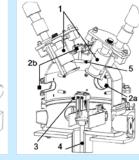


10 17 20 (°) 18 19



Area and FWHM₂₀, of in-situ XRD peaks of sample 11 (a, c) and sample 12 (b, d), versus time $(\Box)(\Box)$ (\Box) T_{1_2} Ni (Deposition/Annealing); $(\bigcirc)'(\bullet)$ {110}B2; (\bullet) {200}B2; $(\triangle)'(\blacktriangle)$ Silicide.





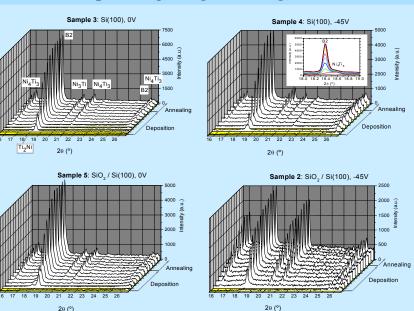
ESRF

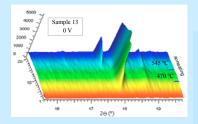
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 -ray windows, lves and lar pumps,
2a. Entrance window for the incoming beam,
2b. Exit window for the diffracted beams,
3. Substrates carrier with heater and bias voltage,
4. Tube to the z drive ,
5. Window protection foils.

Diffractometer of beamline BM20 at the ESRF (ROBL-CRG) and perspective views of the sputter deposition chamber.

Sputtering using Ni-Ti target







This study presents the first reported experiments on in situ analysis of the sputtering of Ni-Ti thin films. These results show that:

- during co-sputtering (Ni-Ti + Ti targets), there is a significant formation of a silicide preceding the growth of the B2 phase;
- also during co-sputtering, the first layers of B2 stack preferentially on {h00} planes;

• on the other hand, during sputtering using a single target (Ni-Ti), the preferential stacking on {110} is present since the beginning of the deposition and there is not such a notorious formation of silicide as it is observed in co-deposition;

• during sputtering using a single target Ni-Ti, there is a significant formation of Ni-rich precipitates; when a bias is applied, there is a much more significant change of the peak intensities of the precipitates Ni_3Ti and Ni_4Ti_3 for the deposition on $SiO_2/Si(100)$, compared to the one on Si(100); • in both cases, there is a significant decrease in intensity of the B2 {110} intensity when a bias (-45 V) is applied;

• also in both cases, during the annealing (at 470°C) after deposition, the intensity of the B2 {110} peak does not increase; for sample 2 this stabilization is reached during the deposition period;

• the FWHM changes significantly during deposition and remains quite stable during intermediate annealing; this trend is more noticeable when no bias is applied.

Acknowledgements

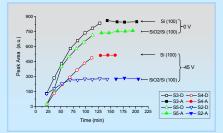
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Evolution along time of the area of B2{110} in-situ Bragg-Brentano diffraction peaks, recorded during: (D): deposition (A): annealing that follows deposition.